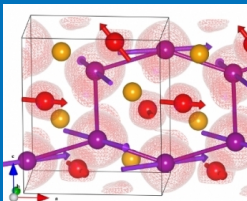
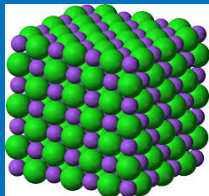


















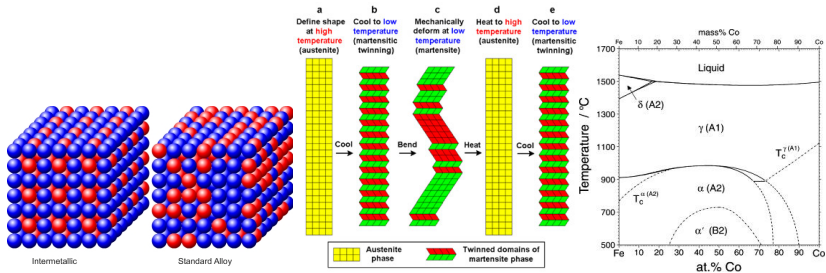
Density functional theory and slowly varying fluctuations at finite temperature to describe alloy and magnetic phase diagrams



 Sc	 Y	 La	 Ce
 Pr	 Nd	 Sm	 Eu
 Gd	 Tb	 Dy	 Ho
 Er	 Tm	 Yb	 Lu

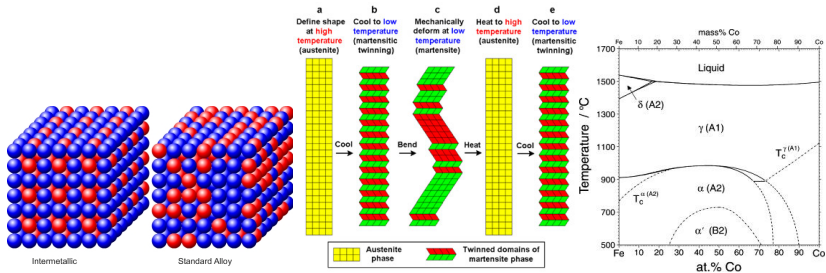
Julie Staunton
University of Warwick

Alloy Solid Solutions and Intermetallics



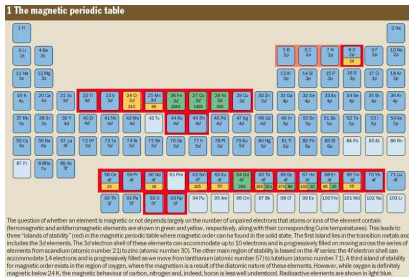
- Mixing of 2 metals (Hume Rothery Rules)
 - band filling (av. no. of electrons/atom).
 - atomic size difference.
 - electronegativity -arrangement of charge around a nucleus.

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- Mixing of 2 metals (Hume Rothery Rules)
 - band filling (av. no. of electrons/atom).
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 - electronegativity -arrangement of charge around a nucleus.
- Solid solution - lattice with each site with $x(1-x)$ chance of A (B) atom on it.
- Ordered arrangement - intermetallic. Affects mechanical,electrical,thermal properties etc.

Magnetic periodic Table

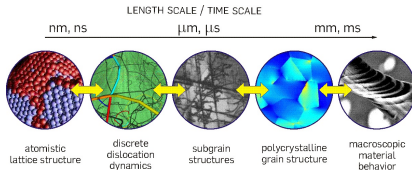
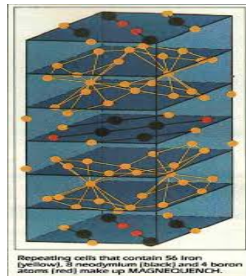
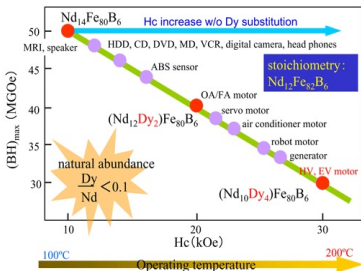


M.Coey and S.Sanvito, Physics World **17**,(11), 34, (Nov. 2004)

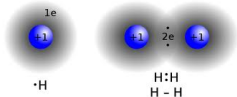
- Transition metal **d**-electrons, high magnetic transition temperatures T_c .
- Rare earth **f**-electrons, high magnetic anisotropy, magnetisation, magnetic coercivity.

Rare earth - transition metal magnets

- Ubiquitous **NdFeB**



- Ab-initio quantum description of electrons and nuclei.
- Simplest many-electron system: H_2 molecule.
- Electrostatic interactions, $H^+ - H^+$, $H^+ - e$, $e - e$.



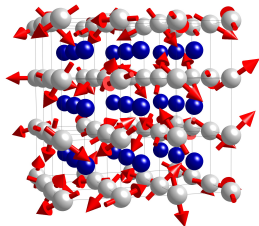
- Assume protons move much more slowly than electrons. Electrons glue protons together.
- Deal with electrons' spins. Pauli Exclusion Principle.
- Good but approximate description. More complex materials??
- Add computational power and tractable method.

Materials Modelling and Density Functional Theory

- $\Omega[\rho, \vec{m}]$, Energy minimised by ground state charge, ρ and magnetisation, \vec{m} , densities.
- Many interacting electrons described in terms of non-interacting electrons in effective fields (Kohn Sham).

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- For magnets these are **local moments**: $\{\hat{e}_i\}$



Finding the interactions amongst the Disordered Local Moments - DLM-DFT

- $P(\{\hat{e}_i\}) = \frac{\exp[-\beta\Omega(\{\hat{e}_i\})]}{\prod_j \int d\hat{e}_j \exp[-\beta\Omega(\{\hat{e}_i\})]}$ where $\Omega(\{\hat{e}_i\})$ is the electronic grand potential from SDFT.

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- Choose 'reference' Hamiltonian $\Omega_0\{\hat{e}_i\}$ and use Feynman Inequality $F \leq F_0 + \langle \Omega - \Omega_0 \rangle^0$ with $\Omega_0 = \sum_i \vec{h}_i \cdot \hat{e}_i$,

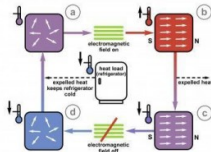
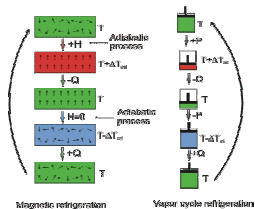
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- 'First-Principles' Mean Field Theory (DLM picture), averaging using techniques adapted from theory for electrons in disordered systems (CPA).

$$P_k(\hat{e}_k) = \frac{\exp[\beta \vec{h}_k \cdot \hat{e}_k]}{\int d\hat{e}_k \exp[-\beta \vec{h}_k \cdot \hat{e}_k]}, \quad \vec{m}_k = \int \hat{e}_k P_k(\hat{e}_k) d\hat{e}_k.$$

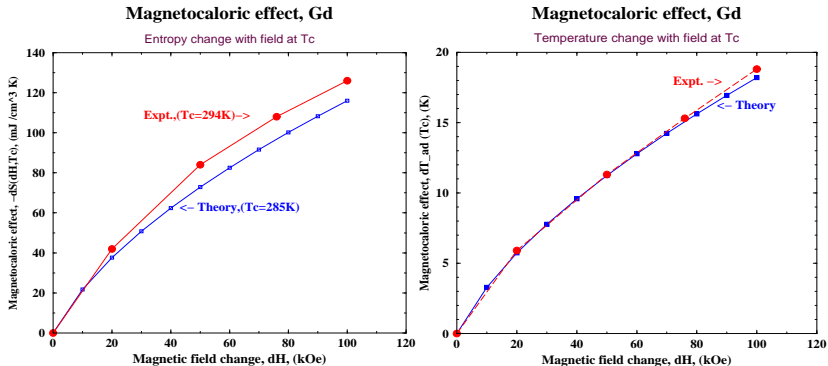
- Free energy: $F(\{\vec{m}_i\}, \vec{H}^{app.}, T) = \bar{\Omega}(\{\vec{m}_i\}, \vec{H}^{app.}) - T \bar{S}_{mag}(\{\vec{m}_i\}, \vec{H}^{app.}) - \vec{H}^{app.} \cdot \sum_i \mu_i \vec{m}_i$

Magnetic Refrigeration - a promising cooling technology



- Magnetic, electronic and lattice entropy interchanging when magnetic field is turned on and off.
- Need materials that respond strongly to modest magnetic fields.
- Large effect when magnetic state changes. Magnetism coupled to another property.
- Rare earth, transition metal magnetic refrigerants.

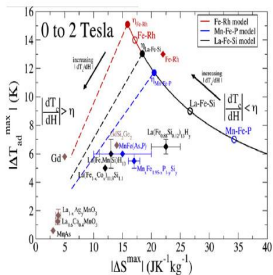
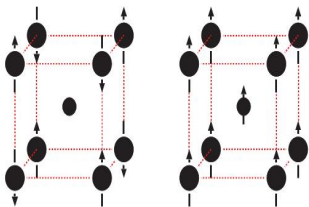
Ab-initio modelling of the MagnetoCaloric Effect in Gadolinium



Experimental results from K.Gschneidner Jr. et al., Rep.Prog.Phys. 68, (2005), 1479-1539

Theory: J.B.Staunton et al, Journal of Physics-Condensed Matter 26, 274210, (2014).

Fe-Rh - a 'two-faced' magnetic alloy



Magnetic states of the ordered B2 (CsCl) alloy **Fe-Rh** and experimental data for $|\Delta T_{ad}^{max}|$ vs. $|\Delta S^{max}|$ for several room temperature magnetic refrigerants, (K. G. Sandeman (Scripta Mat. **67**, 566-571, (2012))).

Metamagnetic transition occurs in tight compositional range, $\text{Fe}_{48}\text{Rh}_{52}$ to $\text{Fe}_{51}\text{Rh}_{49}$, preparation-route dependent, impurities.

FeRh: A little compositional disorder goes a long way . . .

- The $\text{Fe}_{50}\text{Rh}_{50}$ solid solution orders into a B2 alloy at $T \approx 1600\text{K}$. Above $T = 0\text{K}$, the composition is $\text{Fe}_{(100-x)}\text{Rh}_x\text{-Rh}_{(100-x)}\text{Fe}_x$, where $x \neq 0$, the ordering incomplete.
- Away from stoichiometry and where compositional ordering is not complete, there can be Fe atoms on 'Rh' sites.
- Dramatic effect on magnetic properties, phase coexistence and broadening of 1st order transition. For $\text{Fe}_{49}\text{Rh}_{51}$ expt. finds a FM-AF transition at 370K ($T_c = 670\text{K}$) with $|\Delta S^{max}| = 22.5 \text{ J K}^{-1} \text{ Kg}^{-1}$ at 2T.

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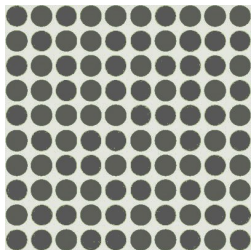
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J.B. Staunton et al. Phys. Rev. B 89, 054427, (2014).

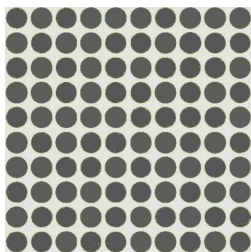
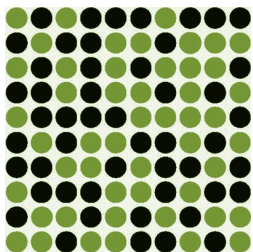
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- Incomplete B2 order: Swapping just **2% of Fe with Rh** causes T_t to drop to 208K ($T_c = 859\text{K}$). At **4%** FM-AF transition has vanished.
- Off stoichiometry: For $\text{Fe}_{96}\text{Rh}_4\text{-Rh}$, $T_t = 549\text{K}$ ($T_c = 700\text{K}$), no transition for $\text{Fe-Rh}_{96}\text{Fe}_4$ ($T_c = 1008\text{K}$).
- For $\text{Fe}_{97}\text{Rh}_3\text{-Rh}_{99}\text{Fe}_1$ ($\text{Fe}_{49}\text{Rh}_{51}$) $T_t = 415\text{K}$ ($T_c = 815\text{K}$). $|\Delta S^{\max}| = 20.7 \text{ J K}^{-1} \text{ Kg}^{-1}$ at 2T.

Electrons and Disorder - An Effective Lattice

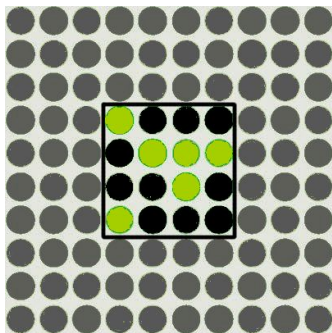


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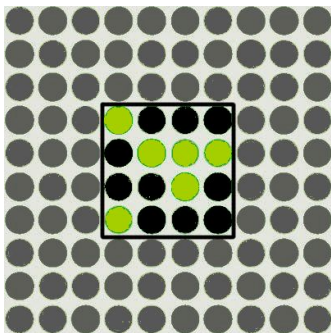


- $\langle G_{ij} \rangle = G_{ij}^0 + \sum_{kl} G_{ik}^0 \Xi_{kl} \langle G_{lj} \rangle$
- $\bar{G}(\mathbf{k}) = \frac{1}{N} \sum_j \langle G_{ij} \rangle e^{i\mathbf{k} \cdot (\mathbf{R}_i - \mathbf{R}_j)} = (G^{0,-1}(\mathbf{k}) - \Xi(\mathbf{k}))^{-1}$
- G_{ii}^η is Green function for embedded impurity.
- $\sum_\eta P(\eta) G_{ii}^\eta = \bar{G}_{ii} \approx \langle G_{ii} \rangle \rightsquigarrow$
Coherent Potential Approximation for Ξ_i .

Cluster Approximation

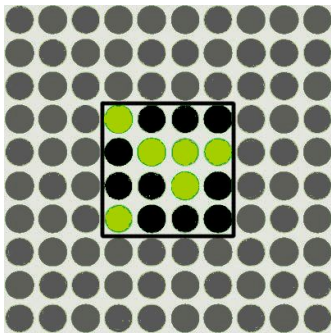


Cluster Approximation



- $G_{IJ}^\eta = [\underline{G}^{0,-1} + \underline{\Xi} - \underline{V}^\eta]_{IJ}^{-1}$
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- $\hat{G}_{IJ} = \frac{1}{\Omega_{BZ}} \sum_{\mathbf{K}_n} \int [G^0(\mathbf{k}) - \Xi(\mathbf{K}_n)]^{-1} e^{i\mathbf{K}_n \cdot (\mathbf{R}_I - \mathbf{R}_J)} d\mathbf{k}_n.$

D.A.Rowlands et al. Phys. Rev. B 67, 115109, (2003); Phys. Rev. B 73, 165122, (2006);

J.B.Staunton et al. JPCM 26, 274210, (2014).

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- Outlook:

- Ab-initio short- and long-range order in alloys, magnets
- Find new adaptive magnetic materials.
- Materials modelling tool for magnetic refrigeration materials, permanent magnets etc..
- Nanostructuring magnetic properties.
- Electronic effects, temperature and spintronics.

Acknowledgements

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Thank you for your attention!